

The Crystal Structure of $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$. A Three-Dimensional Refinement*

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(Received 19 June 1964)

The structure of $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ has been refined by the method of least squares with three-dimensional X-ray diffraction data measured by counter techniques from an approximately spherical crystal. Unit-cell dimensions are $a=5.454$, $b=4.857$, $c=8.173$ Å, and $\beta=107^\circ 22'$, and the space group is $P2_1$. Although a final R of 3.8% was achieved, the data were not accurate enough to locate the hydrogen atoms. Several recently determined sulfate structures are compared and some conclusions drawn.

Introduction

The structure of $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ was originally determined by Ziegler (1934). This structure was further refined by Larson & Helmholz (1954) who used a limited set of data obtained from oscillation photographs. The author now has modern computing facilities and experimental equipment available to him, and, because of continued interest in the compound by other workers (*e.g.* MacGrath, Silvidi & Carroll, 1959; Hirahara & Murakami, 1956), thought it desirable to refine the structure further. Furthermore, this is a nearly ideal compound for the accurate determination of the size and configuration of the sulfate group. This group represents 75% of the scattering matter in the unit cell, and with Mo $K\alpha$ radiation μ is ~ 6.0 cm^{-1} . The small size of the lithium ions may allow them to get close enough to the oxygen atoms to distort the sulfate group; if so, this should give some indication of the maximum distortion to be anticipated in the sulfate group in an ionic crystal.

Experimental

An approximately spherical crystal having an average radius of 0.008 cm was used in collecting intensity data with the single-crystal orienter and Mo $K\alpha$ radiation. The tube was operated at 35 kV and the reflected beam was filtered with 0.006 in. Zr foil. Lattice constants were remeasured and found to be $a=5.454$,

$b=4.857$, $c=8.173$ Å, all ± 0.002 Å and $\beta=107^\circ 22' \pm 2'$ (λ Mo $K\alpha_1=0.70926$ Å). The only extinctions are $0k0$ with k odd. These, together with other experimental data (Ziegler, 1934), indicate that the space group is $P2_1$. Reflections with $\theta_{\text{Mo}} \leq 37.5^\circ$ were examined with a krypton-filled proportional counter, and of 1191 non-extinguished reflections within this hemisphere, 1155 were observed. Lp corrections were applied but because of the nearly spherical shape and low absorption ($\mu R=0.05$) no absorption corrections were necessary. Background corrections were assumed to be dependent on θ only, except for reflections having a higher or lower order within the sphere of measurement. For these reflections an empirical correction was applied, as described in the Appendix, to account for the white radiation.

Refinement of the structure

The structure was refined by full-matrix least-squares minimizing $\sum w(|F_o| - |F_c|)^2$ with $w=1.0$ for all observed reflections and $w=0.0$ for all unobserved reflections. Starting parameters were those of Larson & Helmholz (1954) corrected to eliminate the several typographical errors. Form factors were used in exponential form

$$f = \sum_{i=1}^2 a_i \exp(-b_i s) + c,$$

where $s = \sin^2 \theta / \lambda^2$, with the parameters given by Forsyth & Wells (1959). After a few cycles with isotropic temperature factors the reliability index,

$$R = \sum w ||F_o| - |F_c|| / \sum w |F_o|,$$

* Work performed under the auspices of the U.S. Atomic Energy Commission.

Table 1. Parameters in $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ after isotropic least-squares refinement

Atom	x	y	z	B
S	0.2936 ± 0.0001	0	0.2084 ± 0.0001	0.73 ± 0.01 Å ²
O(1)	0.0207 ± 0.0006	0.0692 ± 0.0007	0.1692 ± 0.0004	1.48 ± 0.04 Å ²
O(2)	0.4364 ± 0.0006	0.1128 ± 0.0007	0.3789 ± 0.0004	1.23 ± 0.04 Å ²
O(3)	0.4008 ± 0.0006	0.1231 ± 0.0007	0.0787 ± 0.0004	1.29 ± 0.04 Å ²
O(4)	0.3301 ± 0.0006	0.7008 ± 0.0007	0.2101 ± 0.0004	1.30 ± 0.04 Å ²
O(w)	0.9104 ± 0.0007	0.4708 ± 0.0011	0.3945 ± 0.0005	2.62 ± 0.07 Å ²
Li(1)	0.3049 ± 0.0013	0.4935 ± 0.0025	0.9940 ± 0.0009	1.45 ± 0.09 Å ²
Li(2)	0.5621 ± 0.0013	0.4862 ± 0.0025	0.3945 ± 0.0009	1.46 ± 0.10 Å ²

Table 2. *Calculated and observed structure factors for $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$*

All F_o 's have been multiplied by 10. Minus signs on F_{obs} should be interpreted as 'less than'. The column headings are h , $|F_{\text{obs}}|$, $|F_{\text{calc}}|$, α .

h	k	l	F_o	$ F_o $	$ F_c $	α															
1	170	187	180	-8	32	30	180														
2	42	17	24	18	24	180	127														
3	278	270	0	-54	55	0	4	106	167	282	-4	57	55	76							
4	31	25	0	-5	13	0	0	5	83	88	-7	41	37	25							
5	94	92	180	0	114	160	0	0	9	87	0	25	0	20							
6	-17	12	180	-3	54	52	180	7	63	61	263	-1	74	74	-2						
7	108	107	0	-2	150	150	0	8	24	30	30	0	52	55	21						
8	54	52	180	-108	170	180	0	1	52	53	251	-1	80	82	39						
9	0	47	46	180	0	38	33	30	0	94	72	101	3	150	145	101					
10	118	121	0	1	118	121	0	0	24	24	83	0	76	76	275						
11	35	35	180	0	9	24	28	83	0	1	13	2	76	75	236						
12	-46	43	180	3	96	98	180	-8	-7	85	206	-5	49	48	74						
13	-8	73	75	0	4	65	63	0	-7	85	206	-5	49	48	74						
14	0	87	86	0	5	23	15	0	-6	22	17	279	-4	26	17	127					
15	-5	31	31	180	0	86	86	180	-2	11	10	10	0	6	6	6					
16	-4	159	158	0	-8	0	0	0	-4	192	193	291	-2	59	58	102					
17	-3	104	100	0	0	47	46	180	-2	113	103	146	-1	39	33	74					
18	-2	426	451	180	-8	67	67	180	-2	113	103	146	-1	39	33	74					
19	-1	364	370	0	-6	18	18	0	-1	42	43	332	0	134	126	234					
20	0	93	105	0	-5	88	88	180	-7	148	148	336	-4	100	100	109					
21	1	444	443	180	-4	50	52	180	7	88	87	108	0	4000	478	40					
22	2	187	185	0	-3	124	119	0	-1	144	140	336	-4	211	210	293					
23	3	209	210	0	-2	80	78	0	4	105	102	109	-3	2	105	100	109				
24	4	103	98	180	-1	130	129	180	5	45	44	75	1	129	124	35					
25	5	27	27	180	0	28	27	0	6	53	50	266	-3	152	145	293					
26	6	107	108	0	1	62	62	0	7	54	54	283	0	30	28	201					
27	7	40	31	180	0	28	27	0	6	18	15	189	1	95	95	357					
28	8	51	52	180	0	28	27	0	6	18	15	189	1	95	95	357					
29	9	0	47	46	180	0	38	33	30	0	7	85	85	206	3	70	74	98			
30	-9	-18	-2	180	0	0	0	0	0	0	0	0	0	5	37	33	363	6	46	46	295
31	-8	-66	-61	0	-7	27	27	0	-8	44	42	270	-8	62	60	199	7	36	33	280	
32	-7	-51	-52	180	-8	-48	-48	180	-5	-122	-126	-284	-7	-59	-57	0	0	0	0	0	
33	-6	-88	-86	180	-7	-62	-61	0	-6	-125	-127	-264	-6	-43	-42	171	-8	-53	-55	140	
34	-5	-117	-118	0	-6	-59	-59	180	-5	-118	-118	-293	-5	-89	-89	259	-4	-76	-76	124	
35	-4	-155	-154	180	-5	-109	-111	180	-4	-125	-135	-352	-4	-53	-53	252	-7	-66	-66	450	
36	-3	-187	-188	0	-4	-135	-135	180	-3	-152	-152	-409	-3	-76	-76	283	-6	-83	-83	374	
37	-2	-261	-261	180	-3	-199	-199	0	-2	-153	-148	-207	-2	-46	-46	303	-5	-63	-63	280	
38	-1	-264	-275	0	-2	-274	-275	0	-1	-180	-181	-63	-1	-97	-97	106	-26	-26	-26	25	
39	0	-248	-258	180	-1	-324	-324	0	0	-2	9	59	59	0	178	178	288	-4	-132	-132	158
40	1	-264	-258	180	0	-127	-125	0	1	129	124	216	0	38	38	208	-1	-52	-52	94	
41	2	-150	-157	0	1	72	75	180	4	68	64	55	0	38	38	208	-1	-52	-52	94	
42	3	-181	-178	180	0	3	3	180	4	59	54	266	0	1	1	101	-16	-16	-16	148	
43	4	-194	-195	180	3	90	102	0	6	74	77	278	4	94	96	204	-2	-76	-76	124	
44	5	-179	-178	0	4	38	33	180	7	26	27	81	5	52	54	188	1	22	13	157	
45	6	-207	-208	180	5	49	49	180	8	24	20	266	6	7	24	20	266	6	7	24	20
46	-6	-17	-6	180	-7	-44	-40	0	-9	-52	-57	-282	-7	-60	-60	133	-8	-44	-44	33	
47	-5	-84	-84	180	-6	-65	-65	0	-8	-65	-65	297	-6	-84	-84	333	-7	-66	-66	283	
48	-4	-116	-116	0	-5	-77	-76	180	-5	-77	-76	256	-5	-81	-81	233	-4	-65	-65	209	
49	-3	-146	-146	180	-4	-93	-93	0	-4	-93	-93	164	-4	-93	-93	164	-3	-85	-85	285	
50	-2	-176	-176	0	-3	-112	-112	180	-3	-112	-112	98	-3	-112	-112	98	-2	-76	-76	124	
51	-1	-208	-208	180	-2	-139	-139	0	-2	-139	-139	202	-2	-139	-139	202	-1	-103	-103	288	
52	0	-240	-240	180	-1	-174	-174	0	-1	-174	-174	262	-1	-174	-174	262	0	-146	-146	380	
53	1	-264	-264	180	0	-210	-210	180	0	-210	-210	332	0	-210	-210	332	1	-180	-180	464	
54	2	-167	-173	180	0	-248	-248	0	1	-112	-113	279	0	1	-112	-113	279	2	-146	-146	380
55	3	-182	-185	180	0	-284	-284	0	2	-150	-145	298	0	2	-150	-145	298	3	-180	-180	464
56	4	-202	-203	180	0	-320	-320	0	3	-202	-202	334	0	3	-202	-202	334	4	-220	-220	560
57	5	-224	-224	180	0	-356	-356	0	4	-284	-284	376	0	4	-284	-284	376	5	-300	-300	760
58	6	-248	-248	180	0	-392	-392	0	5	-356	-356	418	0	5	-356	-356	418	6	-380	-380	960
59	7	-276	-276	180	0	-428	-428	0	6	-428	-428	460	0	6	-428	-428	460	7	-460	-460	1160
60	8	-308	-308	180	0	-464	-464	0	7	-464	-464	492	0	7	-464	-464	492	8	-540	-540	1380
61	9	-344	-344	180	0	-500	-500	0	8	-500	-500	524	0	8	-500	-500	524	9	-620	-620	1560
62	10	-384	-384	180	0	-536	-536	0	9	-536	-536	556	0	9	-536	-536	556	10	-700	-700	1760
63	11	-428	-428	180	0	-572	-572	0	10	-572	-572	588	0	10	-572	-572	588	11	-780	-780	1960
64	12	-476	-476	180	0	-608	-608	0	11	-608	-608	620	0	11	-608	-608	620	12	-860	-860	2160
65	13	-528	-528	180	0	-644	-644	0	12	-644	-644	652	0	12	-644	-644	652	13	-940	-940	2360
66	14	-584	-584	180	0	-680	-680	0	13	-680	-680	684	0	13	-680	-680	684	14	-1020	-1020	2560
67	15	-644	-644	180	0	-716	-716	0	14	-716	-716	716	0	14	-716	-716	716	15	-1100	-1100	2760
68	16	-708	-708	180	0	-752	-752	0	15	-752	-752	748	0	15	-752	-752	748	16	-1180	-1180	2960
69	17	-776	-776	180	0	-788	-788	0	16	-788	-788	780	0	16	-788	-788	780	17	-1260	-1260	3160
70	18	-848	-848	180	0	-824	-824	0	17	-824	-824	812	0	17	-824	-824	812	18	-1340	-1340	3360
71	19	-924	-924	180	0	-860	-860	0	18	-860	-860	844	0	18	-860	-860	844	19	-1420	-1420	3560
72	20	-1004	-1004	180	0	-896	-896	0	19	-896	-896	876	0	19	-896	-896	876	20	-1500	-1500	3760
73	21	-1088	-1088	180	0	-932	-932	0	20	-932	-932	908	0	20	-932	-932	908	21	-1580	-1580	3960
74	22	-1176	-1176	180	0	-968	-968	0	21	-968	-968	940	0	21	-968	-968	940	22	-1660	-1660	4160
75	23	-1268	-1268	180	0	-1004	-1004	0	22	-1004	-1004	972	0	22	-1004	-1004	972	23	-1740	-1740	4360
76	24	-1364	-1364	180	0	-1040	-1040	0	23	-1040	-1040	1004	0	23	-1040	-1040	1004	24	-1820	-1820	4560
77	25	-1464	-1464	180	0	-1076	-1076	0	24	-1076	-1076	1036	0	24	-1076	-1076	1036	25	-1900	-1900	4760
78	26	-1568	-1568	180	0	-1112	-1112	0	25	-1112	-1112	1068	0	25	-1112	-1112	1068	26	-1980	-1980	4960
79	27	-1676	-1676	180	0	-1148	-1148	0	26	-1148	-1148	1100	0	26	-1148	-1148	1100	27	-2060	-2060	5160
80	28	-1788	-1788	180	0	-1184	-1184	0	27	-1184	-1184	1132	0	27	-1184	-1184	1132	28	-2140	-2140	5360
81	29	-1904	-1904	180	0	-1220	-1220	0	28	-1220	-1220	1164	0	28	-1220	-1220	1164	29	-2220	-2220	5560
82	30	-2024	-2024	180	0	-1256</															

dropped to 6.8%, with unobserved reflections omitted. Hydrogen atoms were not included. The final parameters for the isotropic refinement are given in Table 1. Because the origin in the y direction in space group $P2_1$ is arbitrary, y_s was fixed at zero.

It had been hoped initially that the hydrogen atoms could be located. With this end in mind, anisotropic thermal parameters were introduced to account for as much of the electron density as possible before attempting to locate the hydrogen atoms. A few least-squares cycles, including anisotropic thermal parameters for all atoms except Li, were computed and R was reduced to 5.3%. At this point the reflections that had relatively large ΔF 's were remeasured and an extinction correction was applied to all of the data. The extinction correction had the form

$$I_{\text{corr}} = \frac{I_{\text{obs}}}{1 - gI_{\text{obs}}}$$

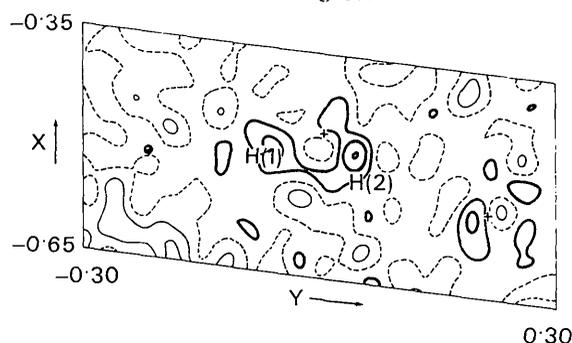


Fig. 1. Difference Fourier section through O(1)-O(w)-O(w). Contour interval is 0.2 e. \AA^{-3} . Heavy lines are positive contours, the dashed line is the zero contour, and the light lines are negative contours. The unit cell here is obtained by the transformation matrix,

$$\begin{pmatrix} 2 & 1 & 0 \\ 0 & 2 & 1 \\ 0 & 0 & 1 \end{pmatrix}$$

and the section is at $z=0.389$ in this cell. + represents O(w) and ● represents O(1).

where $g=8.0 \times 10^{-5}$. A few more least-squares cycles were calculated with anisotropic Li thermal parameters and Dawson's (1960) sulfur form factor in exponential form ($a_1=7.698$, $b_1=1.301$, $a_2=6.820$, $b_2=25.95$, $c=1.477$). The R index was reduced to 3.8%. The sum of the residuals squared was 179 electrons². Thus, $\sigma(\rho_{\text{obs}})=0.26 \text{ e. \AA}^{-3}$ according to the formula of Cruickshank (1950). The final observed and calculated structure factors are given in Table 2. These structure factors were calculated by using the final parameters of Table 3. The experimental values have been corrected for extinction and for the white radiation streak. The ratio of the final changes to their standard deviations was less than 10^{-3} for all parameters.

A three-dimensional ΔF Fourier synthesis was then computed. This showed no peak or hole greater than $\pm 0.6 \text{ e. \AA}^{-3}$. A section through the atoms participating in the probable hydrogen bonding is shown in Fig. 1. The hydrogen atom positions found by Smith (1960) are shown. H(1) is 0.2 \AA below the section and H(2) is 0.1 \AA below the section. The peaks near the hydrogen positions have values of 0.52 and 0.60 e. \AA^{-3} , less than 3 times the estimated $\sigma(\rho)$, and are only about 0.8 and 0.7 \AA from the water oxygen atom. Other peaks and valleys of similar magnitude appeared in the sulfate group. It was concluded that the present set of data is not precise enough to locate the hydrogen atoms in the compound. The water oxygen has a relatively large thermal motion and the hydrogen bonds, as proposed by Larson & Helmholz (1954), are rather long. The hydrogen atoms can be expected to have rather large thermal parameters and thus should be quite difficult to locate by X-ray diffraction techniques.

All interatomic distances less than 3.5 \AA (excluding S-O and S-Li distances $>2.9 \text{ \AA}$) are given in Table 4 along with bond angles about the sulfur atom and O(w). The standard deviations of these distances and angles were calculated using all of the terms from the

Table 3. Position and thermal parameters

Final least-squares position parameters						
Atom	x	y	z			
S	0.2938 ± 0.0001	0	0.2084 ± 0.0001			
O(1)	0.0217 ± 0.0004	0.0701 ± 0.0006	0.1694 ± 0.0003			
O(2)	0.4355 ± 0.0004	0.1129 ± 0.0005	0.3787 ± 0.0002			
O(3)	0.4010 ± 0.0004	0.1230 ± 0.0005	0.0789 ± 0.0003			
O(4)	0.3298 ± 0.0004	0.7008 ± 0.0005	0.2102 ± 0.0003			
O(w)	0.9112 ± 0.0005	0.4718 ± 0.0009	0.3950 ± 0.0004			
Li(1)	0.3053 ± 0.0009	0.4949 ± 0.0018	0.9938 ± 0.0006			
Li(2)	0.5619 ± 0.0010	0.4878 ± 0.0018	0.3950 ± 0.0006			
Final least-squares anisotropic thermal parameters $\times 10^4$						
Atom	B_{11}	B_{22}	B_{33}	B_{12}	B_{13}	B_{23}
S	84 ± 1	81 ± 2	28 ± 1	6 ± 4	25 ± 1	-3 ± 3
O(1)	82 ± 5	241 ± 10	67 ± 3	43 ± 11	29 ± 6	-35 ± 9
O(2)	150 ± 6	133 ± 7	36 ± 2	-49 ± 12	21 ± 6	-15 ± 8
O(3)	163 ± 7	142 ± 7	51 ± 3	55 ± 12	101 ± 7	48 ± 8
O(4)	182 ± 7	96 ± 7	50 ± 3	22 ± 12	28 ± 7	-1 ± 8
O(w)	190 ± 8	334 ± 17	152 ± 5	-63 ± 21	181 ± 10	-103 ± 17
Li(1)	155 ± 14	151 ± 17	60 ± 6	-33 ± 49	52 ± 15	39 ± 34
Li(2)	160 ± 14	156 ± 20	57 ± 6	71 ± 46	36 ± 15	11 ± 29

Table 4. *Interatomic distances in $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$*

No corrections have been made for thermal vibration. Distances from the O(w) atoms at the edges of the Li(2) coordination tetrahedron are marked*.

S-O(1)	1.462 ± 0.002 Å	Li(2)-O(w)	1.907 ± 0.006 Å
O(2)	1.482 ± 0.002	O(2)	1.937 ± 0.009
O(3)	1.479 ± 0.002	O(2)	1.942 ± 0.006
O(4)	1.466 ± 0.002	O(4)	1.953 ± 0.006
O(1)-O(2)	2.400 ± 0.003	O(w)-O(2)	3.091 ± 0.003
O(3)	2.409 ± 0.003	O(2)	3.095 ± 0.004
O(4)	2.412 ± 0.003	O(4)	3.274 ± 0.004
O(2)-O(3)	2.401 ± 0.003	O(2)-O(2)	3.081 ± 0.002
O(4)	2.400 ± 0.003	O(4)	3.148 ± 0.003
O(3)-O(4)	2.399 ± 0.004	O(4)	3.247 ± 0.003
Li(1)-O(1)	1.917 ± 0.005	O(w)-O(w)(2)	2.964 ± 0.003
O(3)	1.950 ± 0.008	O(1)	2.869 ± 0.004
O(3)	1.969 ± 0.006	O(2)	3.091 ± 0.003
O(4)	2.001 ± 0.007	O(2)	3.095 ± 0.004
O(1)-O(3)	3.252 ± 0.003	O(2)	3.385 ± 0.004
O(3)	3.370 ± 0.003	O(4)	3.274 ± 0.004
O(4)	3.187 ± 0.003	O(4)	3.285 ± 0.004
O(3)-O(3)	3.092 ± 0.002	Li(2)	1.907 ± 0.006
O(4)	3.070 ± 0.003		
O(4)	3.155 ± 0.003	O(1)-O(4)	3.462 ± 0.003
O(1)-S-O(2)	109.2 ± 0.1°	O(w)-O(w)-O(w)	110.0 ± 0.2°
O(3)	110.0 ± 0.1	O(1)	73.7 ± 0.2
O(4)	110.9 ± 0.1	O(1)	146.5 ± 0.2
O(2)-S-O(3)	108.4 ± 0.1	O(2)	57.8 ± 0.1
O(4)	109.1 ± 0.1	O(2)	108.1 ± 0.2
O(3)-S-O(4)	109.1 ± 0.1	O(4)	80.6 ± 0.2
		O(4)	113.1 ± 0.2
O(1)-O(w)-O(2)	44.2 ± 0.2	Li(2)	96.9 ± 0.3
O(4)	68.1 ± 0.2	Li(2)	100.8 ± 0.3
Li(2)	115.4 ± 0.4	O(2)-O(w)-O(4)	56.3 ± 0.1
O(4)-O(w)-Li(2)	144.9 ± 0.4	Li(2)	151.3 ± 0.4

Table 5. *Analysis of anisotropic thermal parameters*

The angles are given with respect to the real lattice vectors

Atom	Axis	Root mean square amplitude	B_i	a	b	c
S	1	0.109 Å	0.94 ± 0.02	12.0 ± 6.3°	78.0 ± 6.3°	107.4 ± 2.5°
	2	0.098	0.76 ± 0.02	101.7 ± 6.4	17.9 ± 9.2	99.2 ± 12.3
	3	0.092	0.67 ± 0.02	92.3 ± 3.0	76.8 ± 12.3	19.9 ± 7.0
O(1)	1	0.104	0.86 ± 0.05	18.2 ± 2.8	99.2 ± 2.6	91.5 ± 3.6
	2	0.177	2.47 ± 0.09	75.3 ± 2.1	26.3 ± 4.2	115.0 ± 4.6
	3	0.140	1.55 ± 0.06	100.6 ± 4.0	65.6 ± 4.6	25.1 ± 4.6
O(2)	1	0.153	1.84 ± 0.06	20.6 ± 5.6	110.4 ± 5.8	109.1 ± 3.5
	2	0.125	1.22 ± 0.06	72.1 ± 5.8	28.5 ± 7.3	116.0 ± 7.8
	3	0.101	0.81 ± 0.05	80.2 ± 3.7	71.0 ± 7.2	33.2 ± 6.4
O(3)	1	0.161	2.06 ± 0.06	48.2 ± 3.8	62.7 ± 3.6	68.8 ± 2.9
	2	0.125	1.23 ± 0.05	127.6 ± 6.0	37.9 ± 6.7	75.4 ± 7.6
	3	0.102	0.81 ± 0.05	115.5 ± 5.6	114.1 ± 7.5	26.2 ± 5.5
O(4)	1	0.166	2.17 ± 0.08	10.0 ± 3.3	84.6 ± 2.8	115.6 ± 3.5
	2	0.106	0.89 ± 0.06	95.6 ± 3.5	5.6 ± 4.5	89.1 ± 9.1
	3	0.123	1.19 ± 0.06	81.7 ± 3.6	88.4 ± 10.4	25.7 ± 3.5
O(w)	1	0.141	1.58 ± 0.07	11.5 ± 3.5	92.3 ± 4.8	118.6 ± 2.8
	2	0.182	2.62 ± 0.11	81.8 ± 5.1	34.4 ± 4.4	61.3 ± 5.2
	3	0.233	4.30 ± 0.12	82.0 ± 2.0	124.3 ± 4.5	42.7 ± 3.7
Li(1)	1	0.141	1.57 ± 0.12	50.4 ± 25.1	72.9 ± 19.9	60.7 ± 22.6
	2	0.117	1.09 ± 0.14	72.4 ± 12.3	41.0 ± 14.9	130.3 ± 16.4
	3	0.157	1.94 ± 0.16	135.1 ± 26.2	54.1 ± 17.8	54.2 ± 12.9
Li(2)	1	0.162	2.06 ± 0.17	30.6 ± 15.6	59.9 ± 17.3	110.2 ± 16.3
	2	0.122	1.17 ± 0.13	117.6 ± 14.6	46.2 ± 24.6	113.0 ± 31.3
	3	0.136	1.47 ± 0.15	102.3 ± 26.6	58.7 ± 25.3	31.4 ± 23.7

Table 6. *Translational and torsional vibration matrices for the sulfate group relative to the axes a, b, c.*

The r.m.s. $\Delta U_{ij} = 0.0015 \text{ \AA}^2$.

$$\tau = \begin{pmatrix} 0.0116 & 0.0005 & -0.0004 \\ & 0.0108 & -0.0001 \\ & & 0.0099 \end{pmatrix} \text{ \AA}^2$$

$$\omega = \begin{pmatrix} 8.7 & -2.8 & -2.2 \\ & 18.4 & 0.7 \\ & & 25.7 \end{pmatrix} \text{ deg}^2$$

$$\sigma\tau = \begin{pmatrix} 0.0009 & 0.0008 & 0.0008 \\ & 0.0009 & 0.0008 \\ & & 0.0005 \end{pmatrix} \text{ \AA}^2$$

$$\sigma\omega = \begin{pmatrix} 2.3 & 2.3 & 2.0 \\ & 2.3 & 2.1 \\ & & 2.3 \end{pmatrix} \text{ deg}^2$$

variance-covariance matrix from the least-squares calculation and neglecting unit-cell errors. Inclusion of unit-cell errors increases these standard deviations by about 20 to 50% depending on the length of the bond.

The anisotropic thermal parameters were analyzed to find the axes of the ellipsoids and their orientation in the unit cell. These results are presented in Table 5. The standard deviations presented in Table 5 were computed using numerical derivatives and the complete variance-covariance matrix. The minor axes of the oxygen make angles of 5.5° , 4.9° , 12.5° and 2.0° with the S-O bonds. Rigid body analysis of the sulfate group was carried out by the method of Cruickshank (1956*a*) using a code supplied by Trueblood (1962). The τ and ω matrices given in Table 6 were calculated with a weight of two for the sulfur atom and one for each of the oxygen atoms. The thermal vibration of the oxygen atoms independent of the sulfur atom was computed from the matrices $B_{O_i} - B_S$ (essentially the in phase assumption of Busing & Levy (1964)). These matrices were diagonalized and the angle of the smallest axis of each with the S-O bond was calculated. These angles were 5.8° , 3.7° , 6.0° and 3.6° . The residual mean square amplitudes along the bonds are -0.0006 , 0.0015 , 0.0003 , and 0.0017 \AA^2 . None of these values is significantly different from zero as is anticipated for a tightly bound group such as the sulfate ion. Corrections to the bond lengths due to thermal motion in the sulfate group were calculated by the method of Cruickshank (1956*b*; 1961) and by the assumption of in phase motion (Busing & Levy, 1964). The results were almost identical and the values from the in phase assumption are presented in Table 7. Also listed in Table 7 are the Li-O distances corrected for this lengthening of the S-O distances*.

Table 7. *Interatomic distances corrected for thermal motion*

S-O(1)	1.473	O(2)-Li(2)	1.934
S-O(2)	1.487	O(3)-Li(1)	1.964
S-O(3)	1.486	O(3)-Li(1)	1.946
S-O(4)	1.473	O(4)-Li(1)	1.998
O(1)-Li(1)	1.908	O(4)-Li(2)	1.949
O(2)-Li(2)	1.939		

The residual oxygen thermal motion was analyzed as rigid body motion. The ω matrix was essentially

* The correction to the Li-O distance was computed from the equation $d_{cor} = d_{obs} + \Delta s_{-o} \cos \alpha$, where Δs_{-o} is the change in the S-O bond length and α is the S-O-Li angle. This gives an estimate of the minimum possible length of this bond.

identical with that from the complete set of data and the τ matrix was essentially zero. The r.m.s. $\Delta U_{i,j}$ was 0.0018 \AA^2 , indicating that the sulfate anion acts like a rigid body and that most of the residual motion of oxygen atoms is due torsional oscillation of the group about the sulfur.

Discussion

A careful analysis of the present data has not resulted in locating the hydrogen atoms in $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$. The probable reason for this is the very high thermal motion of these atoms. Their location has been determined by Smith (1960) using neutron diffraction techniques. He found them to have thermal parameters of about 7.0. The positions are the same as proposed by Larson & Helmholtz (1954). Other evidence of high thermal motion for the hydrogen atoms is given by the Raman spectra (Vassas-Dubuisson, 1953) from which it is found that the O-H stretching frequencies are nearly the same as in water vapor.

A projection of the structure on the *ac* plane is given in Fig. 2. It can be seen that the two Li ions have very similar environments. They both have four oxygen neighbours at an average distance of 1.947 \AA uncorrected for the thermal motion of the sulfate ion and 1.943 \AA corrected for the thermal motion. The range (corrected) is from 1.907 \AA for O(w)-Li(2) to 1.998 \AA for O(4)-Li(1). The two short distances are to the water oxygen and to O(1), the sulfate oxygen involved in the weak hydrogen bonding. The other six Li-O distances are to oxygen atoms which have two Li-O contacts as well as the S-O bond.

The crystal structures of several inorganic sulfate compounds have been determined with quite high precision during the last few years. Pertinent data relating to the sulfate group in several of these compounds are given in Table 8. All corrections to the S-O distances have been calculated in the same manner, *i.e.* that described above for $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$. The average values of isotropic B_O were calculated from the anisotropic refinement (Hamilton, 1959). The average uncorrected S-O distances are nearly identical, all being within twice the estimated error of the most precisely known, $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$. The average corrected distances are in two groups, neither of which is significantly different from the other, but this may be a real effect. We note that in the three compounds with the longer corrected distance the sulfate group is surrounded by hydrogen atoms only, and in the other two compounds the cations

Table 9. Comparison of 00*l* reflections with and without the streak correction

Data have been corrected for extinction.

<i>l</i>	<i>I</i> _{Meas}	With streak correction		Without streak correction	
		<i>F</i> _o	<i>F</i> _c	<i>F</i> _o	<i>F</i> _c
1	1154	9.3	10.5	9.3	10.5
2	3965	26.8	25.9	26.2	25.6
3	1267	16.7	17.3	17.2	17.7
4	269	8.7	8.2	9.4	8.2
5	2819	35.8	36.1	35.2	35.9
6	103	3.1	3.3	6.9	3.6
7	629	19.0	19.4	19.4	19.7
8	63	4.7	4.6	6.5	4.9
9	25	2.8	2.7	4.3	2.8
10	174	12.7	12.5	12.9	12.8
11	10	< 2.2	0.5	3.0	0.8
12	40	6.6	7.1	6.9	7.1
13	7	1.9	2.0	2.6	2.4

tors, to give a measure of the effective diffracted power at each value of 2θ as seen through a window of width $\Delta(2\theta)$. Because the range of wave lengths seen through this window is a function of θ , we must correct for this by dividing the data by an effective window width parameter ($\csc \theta$). This gives a measure of the effective diffracted power of each wavelength seen by the counter system. By taking the ratio of the effective diffracted power at the $K\alpha$ line to that at any other point, one can determine how much intensity a reflection will contribute to the intensity at the other point along the lattice row. In order to obtain this information, one ordinarily must take lattice rows with several observed reflections. The choice of lattice rows to look at should be made from those in which the first line is the strongest and the second line quite a bit weaker, so that its contribution to the first line can be ignored. One then measures the intensity along the lattice row and processes it using equation (1) starting with the small 2θ values:

$$R_{\lambda} = \frac{\frac{G(\theta)}{\csc \theta} - \sum_{k=2}^K \{G(k) R_{\lambda_k}\}}{G(1)} \quad (1)$$

where $G(\theta)$ is the intensity at θ divided by the Lorentz and polarization factors

$G(k)$ is $G(\theta)$ at the k th reflection

R_{λ_k} is the ratio of the effective diffracted power of the wave length λ_k which reflects at θ to the effective diffracted power at the k th reflection.

The apparent inconsistency at the first reflection, where R_{λ_1} should be equal to 1.0 rather than $\sin \theta$, is removed when one remembers that the $\csc \theta$ term corrects for ranges of λ in the white radiation streak. In general the intensity of the $K\alpha$ line is many times stronger than the white radiation streak, and hence at the first reflection only one wave length ($K\alpha$) is seen rather than a range of wave lengths as at other points along the white radiation streak. The observed intensity data can then be processed with equation (2) to obtain a set of corrected relative $|F|^2$ values:

$$|F_{nH}|^2 = \{I_{nH} - [\sum_i (I_{iH} R_{\lambda_i})] \csc \theta\} / L_p \quad (2)$$

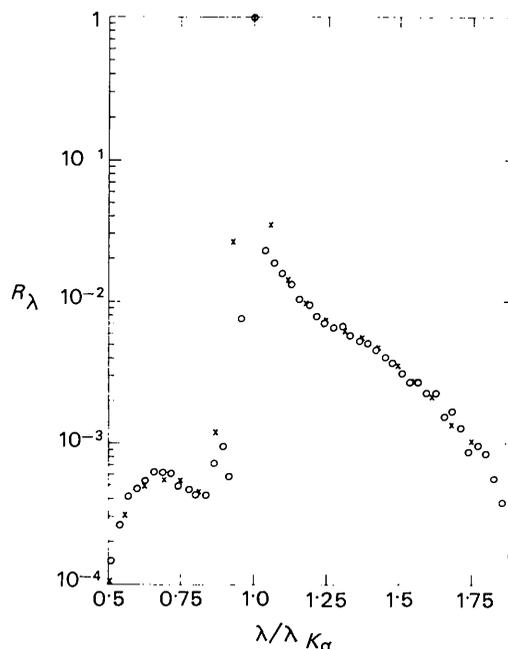
where I_{nH} is the observed intensity of the n th reflection along the lattice row H

R_{λ_i} is as defined in equation (1)

L_p is the Lorentz-polarization factor

\sum_i denotes the sum over all reflections along the lattice row except the n th.

In $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ the $0k0$ and $h0h$ lattice rows fit the above conditions, and since the first reflections occur at quite different 2θ values (16.84° for 020 and 8.02° for $\bar{1}01$) it was felt that processing the two data sets would give a fairly good test of the correctness of the assumptions of equation (1). Values of R_{λ} obtained from both sets of data are presented in Fig. 3. The circles represent the $0k0$ data and the crosses the $h0h$ data. The intensity data for $\text{Li}_2\text{SO}_4 \cdot \text{H}_2\text{O}$ were processed with and without the above correction. The

Fig. 3. Plot of R_{λ} vs. $\lambda/\lambda_{K\alpha}$.

final R values were 4.0% and 4.5% respectively and the standard deviations of the S–O bonds were 0.002 Å and 0.003 Å (the value of g was 6.4×10^{-5}). A perhaps more dramatic effect of this correction can be seen by examination of weak reflections along a lattice row containing some quite strong reflections. Table 9 presents the data for the 00/ lattice row. The agreement of the 006 and 0011 reflections can be seen to be much improved after correcting for the white radiation.

I wish to thank Drs. Don T. Cromer, R. B. Roof, Jr., and K. N. Trueblood for helpful discussions and Drs. W. H. Baur and H. Smith for making the results of their work available to me prior to publication.

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Mixed Oxides of Titanium and Niobium: The Crystal Structure of $\text{TiNb}_{24}\text{O}_{62}$ ($\text{TiO}_2 \cdot 12\text{Nb}_2\text{O}_5$)

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(Received 1 June 1964)

$\text{TiNb}_{24}\text{O}_{62}$ is monoclinic, with $a = 29.78$, $b = 3.821$, $c = 21.12$ Å, $\beta = 94.9^\circ$, space group $C2$. The structure, refined by two-dimensional Fourier methods, contains ReO_3 -type blocks of metal–oxygen octahedra, 3×4 and infinite in extension along b . These join up in pairs, and the double blocks pack together by additional edge sharing, with metals in tetrahedral coordination at each of the junctions. The relationship to the structure of high-temperature Nb_2O_5 , as well as to other known and hypothetical compounds formed from blocks of the same size, is examined.

Introduction

The phase equilibrium relations, and the crystal structures of the phases in the system TiO_2 – Nb_2O_5 , have been independently studied by the present authors (Roth & Coughanour, 1955; Wadsley, 1961a, b). Both of us concluded that only two compounds existed, TiNb_2O_7 ($\text{TiO}_2 \cdot \text{Nb}_2\text{O}_5$) and a second identified by structure-analysis as $\text{Ti}_2\text{Nb}_{10}\text{O}_{29}$ ($2\text{TiO}_2 \cdot 5\text{Nb}_2\text{O}_5$) occurring in two forms, orthorhombic and monoclinic. These are closely related to each other, and can be described as the members $n = 3$ and 4 of a ‘homologous series’ of oxides with the general formula $B_{3n}\text{O}_{8n-3}$, their structures being derived from that of a parent compound $B_3\text{O}_8$, at the time unknown, but now

reported by Andersson (1964) for the oxyfluoride $\text{Nb}_3\text{O}_7\text{F}$. Attempts to prepare the additional homologue $n = 5$ ($\text{TiNb}_{14}\text{O}_{37}$, or $\text{TiO}_2 \cdot 7\text{Nb}_2\text{O}_5$), and to identify the member $n = 6$ ($\text{Nb}_{18}\text{O}_{45}$) with one of the polymorphs of Nb_2O_5 , were unsuccessful (Wadsley, 1961b). This particular series of oxides appeared to contain only the two compounds, unlike the reduced tungsten-molybdenum oxides $(\text{W}, \text{Mo})_n\text{O}_{3n-1}$ (Magnéli, 1953) and the titanium oxides $\text{Ti}_n\text{O}_{2n-1}$ (Andersson & Jahnberg, 1963) where both have six or more members each related by a common structural principle. Likewise Kihlberg (1963) found only three members of a series $\text{Mo}_n\text{O}_{3n-m+1}$.

The high temperature form of Nb_2O_5 , and the reduced phase $\text{Nb}_{22}\text{O}_{54}$ found by Norin & Magnéli (1961), can be grouped together by another series-type general formula $B_{3n+1}\text{O}_{8n-2}$; Nb_2O_5 corresponds to $n = 9$ and $\text{Nb}_{22}\text{O}_{54}$ to $n = 7$ (Gatehouse & Wadsley, 1964). A new

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